Nanostructured Materials and Nanotechnology VII

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Nanostructured Materials and Nanotechnology VII

Nanostructured Materials and Nanotechnology VII

A Collection of Papers Presented at the 37th International Conference on Advanced Ceramics and Composites January 27–February 1, 2013 Daytona Beach, Florida

> Edited by Sanjay Mathur Francisco Hernandez-Ramirez

> > Volume Editors Soshu Kirihara Sujanto Widjaja



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Preface

This CESP issue contains papers that were presented during two symposia held during the 37th International Conference and Exposition on Advanced Ceramics and Composites, Daytona Beach, Florida, January 27–February 1, 2013:

- Symposium 7: 7th International Symposium on Nanostructured Materials and Nanocomposites
- Focused Session 3: Nanomaterials for Sensing Applications: Fundamental Material Designs to Device Integration

Over 90 contributions (invited talks, oral presentations, and posters) were presented by participants from universities, research institutions, and industry, which offered interdisciplinary discussions indicating strong scientific and technological interest in the field of nanostructured systems. This issue contains 15 peer-reviewed papers cover various aspects and the latest developments related to nanoscaled materials.

The editors wish to extend their gratitude and appreciation to all the authors for their cooperation and contributions, to all the participants and session chairs for their time and efforts, and to all the reviewers for their valuable comments and suggestions. Financial support from the Engineering Ceramic Division of The American Ceramic Society (ACerS) is gratefully acknowledged. The invaluable assistance of the ACerS staff of the meetings and publication departments, instrumental in the success of the symposium, is gratefully acknowledged.

We believe that this issue will serve as a useful reference for the researchers and technologists interested in science and technology of nanostructured materials and devices. SANJAY MATHUR
University of Cologne, Germany
FRANCISCO HERNANDEZ-RAMIREZ
Catalonia Institute for Energy Research and University of Barcelona, Spain

Introduction

This issue of the Ceramic Engineering and Science Proceedings (CESP) is one of nine issues that has been published based on manuscripts submitted and approved for the proceedings of the 37th International Conference on Advanced Ceramics and Composites (ICACC), held January 27–February 1, 2013 in Daytona Beach, Florida. ICACC is the most prominent international meeting in the area of advanced structural, functional, and nanoscopic ceramics, composites, and other emerging ceramic materials and technologies. This prestigious conference has been organized by The American Ceramic Society's (ACerS) Engineering Ceramics Division (ECD) since 1977.

The 37th ICACC hosted more than 1,000 attendees from 40 countries and approximately 800 presentations. The topics ranged from ceramic nanomaterials to structural reliability of ceramic components which demonstrated the linkage between materials science developments at the atomic level and macro level structural applications. Papers addressed material, model, and component development and investigated the interrelations between the processing, properties, and microstructure of ceramic materials.

The conference was organized into the following 19 symposia and sessions:

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Symposium Mechanical Behavior and Performance of Ceramics and Composites

Symposium Advanced Ceramic Coatings for Structural, Environmental, and
Functional Applications

Symposium 10th International Symposium on Solid Oxide Fuel Cells (SOFC):
Materials, Science, and Technology

Symposium Armor Ceramics

Symposium Next Generation Bioceramics
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Symposium 6	International Symposium on Ceramics for Electric Energy Generation, Storage, and Distribution
Symposium 7	7th International Symposium on Nanostructured Materials and Nanocomposites: Development and Applications
Symposium 8	7th International Symposium on Advanced Processing & Manufacturing Technologies for Structural & Multifunctional Materials and Systems (APMT)
Symposium 9	Porous Ceramics: Novel Developments and Applications
Symposium 10	Virtual Materials (Computational) Design and Ceramic Genome
Symposium 11	Next Generation Technologies for Innovative Surface Coatings
Symposium 12	Materials for Extreme Environments: Ultrahigh Temperature Ceramics (UHTCs) and Nanolaminated Ternary Carbides and Nitrides (MAX Phases)
Symposium 13	Advanced Ceramics and Composites for Sustainable Nuclear Energy and Fusion Energy
Focused Session 1	Geopolymers and Chemically Bonded Ceramics
Focused Session 2	Thermal Management Materials and Technologies
Focused Session 3	Nanomaterials for Sensing Applications
Focused Session 4	Advanced Ceramic Materials and Processing for Photonics and Energy
Special Session	Engineering Ceramics Summit of the Americas
Special Session	2nd Global Young Investigators Forum

The proceedings papers from this conference are published in the below nine issues of the 2013 CESP; Volume 34, Issues 2-10:

- Mechanical Properties and Performance of Engineering Ceramics and Composites VIII, CESP Volume 34, Issue 2 (includes papers from Symposium 1)
- Advanced Ceramic Coatings and Materials for Extreme Environments III, Volume 34, Issue 3 (includes papers from Symposia 2 and 11)

- Advances in Solid Oxide Fuel Cells IX, CESP Volume 34, Issue 4 (includes papers from Symposium 3)
- Advances in Ceramic Armor IX, CESP Volume 34, Issue 5 (includes papers from Symposium 4)
- Advances in Bioceramics and Porous Ceramics VI, CESP Volume 34, Issue 6 (includes papers from Symposia 5 and 9)
- Nanostructured Materials and Nanotechnology VII, CESP Volume 34, Issue 7 (includes papers from Symposium 7 and FS3)
- Advanced Processing and Manufacturing Technologies for Structural and Multi functional Materials VII, CESP Volume 34, Issue 8 (includes papers from Symposium 8)
- Ceramic Materials for Energy Applications III, CESP Volume 34, Issue 9 (includes papers from Symposia 6, 13, and FS4)
- Developments in Strategic Materials and Computational Design IV, CESP Volume 34, Issue 10 (includes papers from Symposium 10 and 12 and from Focused Sessions 1 and 2)

The organization of the Daytona Beach meeting and the publication of these proceedings were possible thanks to the professional staff of ACerS and the tireless dedication of many ECD members. We would especially like to express our sincere thanks to the symposia organizers, session chairs, presenters and conference attendees, for their efforts and enthusiastic participation in the vibrant and cutting-edge conference.

ACerS and the ECD invite you to attend the 38th International Conference on Advanced Ceramics and Composites (http://www.ceramics.org/daytona2014) January 26–31, 2014 in Daytona Beach, Florida.

To purchase additional CESP issues as well as other ceramic publications, visit the ACerS-Wiley Publications home page at www.wiley.com/go/ceramics.

SOSHU KIRIHARA, *Osaka University, Japan* SUJANTO WIDJAJA, *Corning Incorporated, USA*

Volume Editors August 2013

Nanostructured Materials and Nanotechnology

SOL-GEL APPROACH TO THE CALCIUM PHOSPHATE NANOCOMPOSITES

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ABSTRACT

The sol-gel chemistry route has been developed to prepare Ca-O-P gels samples. In the sol-gel process 1,2-ethanediol and EDTA were used as complexing agents. Additionally, triethanolamine and polyvinyl alcohol were used as gel formina materials. network Calcium phosphate/hydroxyapatite thin films were obtained on substrate by dip-coating technique. The final silicon nanocomposites were obtained by calcination of coatings for different time at 1000°C. It was shown that adjustment of heating time and dip-coating conditions can be used to control the synthesis processing, phase purity and morphology of thin films. It was concluded, that the formation of calcium phosphate/hydroxyapatite composites in some cases is promoted by dipping time.

INTRODUCTION

Calcium hydroxyapatite (CHA) coatings have received considerable attention because they exhibit bone bonding capabilities, i.e. excellent biocompatibility, bioactivity and osteoconductivity). 1,2,3 CHA coatings on different substrates (Ti-6A1-4V alloy, NiTi alloy, Mg, Ti, Si, steel) are being widely used in orthopedics and dentistry. 4,5,6,7,8,9 Many preparation techniques are used currently in coating CHA onto different substrates. However, some metastable and amorphous phases appear in the CHA coating during plasma spraying process 10,11 or pulsed laser the ${\rm deposition}^{12}$ which result in the low crystallinity of CHA coating. The biomimetic CHA coatings have the limitation of poor adhesion and lower growth rates. 13,14 The sol-gel and hydrothermal methods are cost effective, low temperature routes for coating hydroxyapatite on various substrates. 3,15 Sol-gel processing also provides a convenient method for applying tricalcium phosphate (TCP) films. 16,17 Calcium phosphate ceramic is well known for its osteoinductive properties, good degradability, high hydrophilicity. 18,19,20 Calcium phosphate cements have been used in medical and dental applications for many years. 18 For example, tetracalcium phosphate is one of the major powder of self-setting orthopedic components and dental cements.^{21,22} However, the low strength and high brittleness of calcium phosphate cements prohibit their use in many stress-bearing locations, which would require an improvement in mechanical properties.²³ It was shown that gelatine addition to calcium phosphate bone cement improves its mechanical properties.²⁴

Calcium phosphate ceramics, which are commonly used as implants for bone reconstruction, appear to be good candidates for biocompatible drug carriers, since they can be resorbed by cells and they promote new bone formation by releasing calcium and phosphate ions.²⁵ Drug-loaded polymers and calcium phosphate composites were also tested as cell and drug carrier materials. 26,27 Recently calcium phosphate systems, including both hydroxyapatite and tricalcium phosphates (CHA-TCP), have attracted significant interest as drug delivery vehicles. It was demonstrated that protein loading and release behaviour of CHA-TCP can be controlled by tailoring particle size and surface area.²⁸ The CHA-TCP cement was suggested as carrier for different drugs, proteins and chemotherapeutic agents. 29,30 Many preparation techniques were suggested for the preparation of CHA-TCP films coating, such as microplasma spray³¹, high-power ion beam ablation sputtering^{33,34} plasma³². rf-magnetron electrochemical/hydrothermal method.³⁵ The approach was used only for the preparation of biphasic CHA-TCP powders. 36,37 In this paper we report on the synthesis and characterization of CHA-TCP thin films on the silicon 33 substrate using dip-coating technique. For the preparation of stable sols a novel sol-gel synthesis approach was suggested.

EXPERIMENTAL

Aqueous sol-gel chemistry route based on phosphoric acid as the phosphorus precursor and calcium acetate monohydrate as source of calcium ions have been developed to prepare Ca-O-P gel samples. These gels were

used as precursors for the deposition of Ca₁₀(PO₄)6(OH)₂-Ca₃(PO₄)₂ (CHA-TCP) composites onto commercial silicon (Si, 1.5×1.5) substrates by dip-coating technique from the Ca-O-P gels stabilized with complexing reagents. In the solgel process, 2.6425 g of calcium acetate monohydrate, Ca(CH₃COO)₂ · H₂O (99.9%; Fluka) was dissolved in 50 ml of distilled water under continuous stirring at 65°C. To this solution 4.82185 g of ethylenediaminetetraacetic acid (EDTA; 99.0%; Alfa Aesar) was added. After stirring at 60-65°C for 1 h, 2 ml of 1,2-ethanediol (99.0%; Alfa Aesar) and 9 ml of triethanolamine (99.0%; Merck) were slowly poured to the solution. After stirring at 60-65°C for 10 h, appropriate amount of phosphoric acid, H₃PO₄ (85.0%; Reachem) was added to the above solution. Finally, 10 ml of 3% polyvinyl alcohol (PVA7200, 99.5%; Aldrich) solution was added. The obtained solution was stirred in a beaker covered with watch glass for 2 h at the same temperature and was used for coating of silicon substrates. Dip-coating method was employed to produce sol-gel coatings. 38,39 The standard immersing (85 mm/min) and withdrawal rates (40 mm/min) for dip-coating process were applied for all the samples. The dipping procedure was repeatedly performed 5, 15 and 30 times. After evaporation of solvent the substrates were dried in an oven for 10 min at 110°C and heated at 1000°C for 5 h with heating rate of 1°C/min.

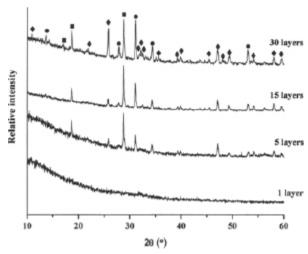
For the characterization of surface properties, the X-ray powder diffraction (XRD) analysis, scanning electron microscopy (SEM), Raman spectroscopy, atomic force microscopy (AFM) and the contact angle measurements were recorded. XRD analysis was performed on a Bruker AXE D8 Focus diffractometer with a LynxEye detector using Cu K_{α} radiation. The measurements were recorded at the standard rate of 1.5–20/min. The scanning electron microscope JEOL JSM 8404 and atomic force microscope

Veeco Bioscope 2 were used to study the surface morphology and microstructure of the obtained thin films. For the characterization of surface hydrophobicity of coatings, the measurements of a contact angle on dipcoating apparatus KVS Instrument CAM 100 were performed. A micro-droplet of water (volume 6 μ l) was allowed to fall onto the sample from a syringe tip to produce a sessile drop. The Raman spectra were registered with confocal Raman spectrometer/microscope LabRam HR 800 using 632.8 nm laser for excitation.

RESULTS AND DISCUSSION

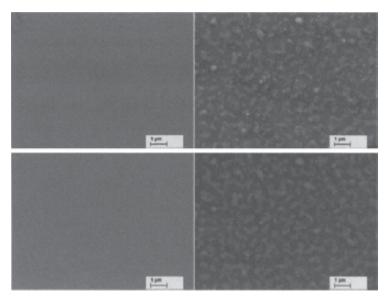
Fig. 1 represents the XRD patterns of films obtained from Ca-O-P gel using dip-coating technique. These results present the influence of the number of coating procedures on the crystallization of calcium phosphate coatings. As seen from Fig. 1, after first immersing, withdrawal and annealing procedure no peaks attributable the $Ca_{10}(PO_{4})_{6}(OH)_{2}$ or $Ca_3(PO_4)_2$ crystal phases are observed. The layer formed contains only amorphous materials. However, already after five dipping and annealing times the main characteristic peaks attributable tricalcium phosphate Ca₃(PO₄)₂ and dicalcium diphosphate Ca₂P₂O₇ (DCDP) crystal phases appear in the XRD pattern. The repetition of immersing, withdrawal and annealing procedures for 15 times did not change phase composition of coating. However, such repeating increased the crystallinity of phosphates significantly since the diffraction lines became more sharp and intense. Finally, with further increasing of calcium phosphate layers up to 30, the formation of calcium hydroxyapatite is evident (diffraction lines of CHA are marked as solid rhombus). 11 The Ca₃(PO₄)₂ and Ca₂P₂O₇ phases also remain in the sample obtained after 30 immersing and annealing procedures. Thus, suggested sol-gel chemistry route could be successfully used for the preparation of CHA-TCP coatings containing dicalcium diphosphate onto silicon substrate.

Figure 1. XRD patterns of the Ca-O-P gel samples annealed at 1000° C after each dipping procedure for 5 h in air. Diffraction lines are marked: \bullet - Ca₁₀(PO₄)₆(OH)₂, \bullet - Ca₃(PO₄)₂ and \bullet - Ca₂P₂O₇.



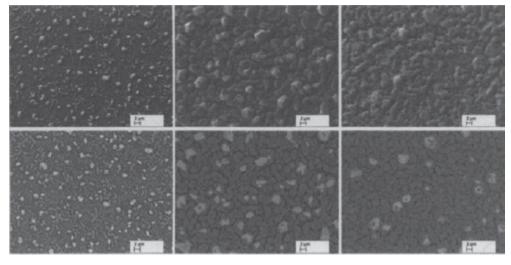
The textural properties of the synthesized samples were investigated by scanning electron microscopy (SEM). Fig. 2 shows SEM micrographs (secondary electron (SE) and back scattered electron (BSE) images) of pure silicon substrate and sample obtained after first immersing, withdrawal and annealing procedure calcined at 1000°C.

<u>Figure 2.</u> SEM micrographs of silicon substrate (at left) and sample containing 1 layer of Ca-O-P gel calcined at 1000°C (at right) in SE (at top) and BSE (at bottom) modes.



The brightness of the silicon substrate on BSE image is highly homogeneous over the entire measuring area. Moreover, the SEM micrographs clearly show that already first layer contains Ca-O-P intermediate amorphous products which consist of differently shaped particles. The additional homogenization of the intermediates and further sol-gel processing are necessary to get CHA-TCP. The SEM micrographs of other three samples are presented in Fig. 3.

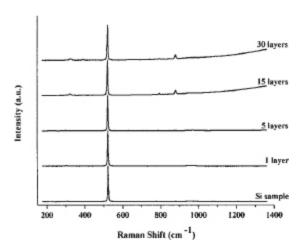
<u>Figure 3.</u> SEM micrographs of sample containing 5 layers (at left), 15 layers (at middle) and 30 layers (at right) of Ca-O-P gel calcined at 1000°C in SE (at top) and BSE (at bottom) modes.



A progressive change in morphology of specimens is evident with increased immersing time. The formation of differently shaped crystallites (spherical particles and platelike grains) with an average grain size ranging between 1 and 2 µm is evident from these investigations. According to the SEM micrographs presented in Fig. 3 the coatings of 15 and 30 layers have similar structural characteristics. There are no macro cracks or pores. However, the amount of particles slightly decreases with spherical increasing amount of the layers on the substrate. Finally, the micrographs of Ca-O-P gel calcined at 1000°C show highly uniform and crystalline particles with smooth surfaces. Therefore, the proposed sol-gel technique appears to be very attractive way to make a high density, homogeneous CHA-TCP ceramic composites. The BSE images clearly demonstrate that most of the material is finely divided, however, the distribution of its chemical elements is not uniform. The formation of multiphasic system composed of 3 different phases is evident. Such observations partially support previous results obtained by XRD analysis. Fig. 1 clearly shows the formation of Ca₁₀(PO₄)₆(OH)₂, Ca₃(PO₄)₂ and Ca₂P₂O₇ crystalline phases only in the sample obtained after 30 immersing and annealing procedures.

On the other hand, the negligible $v_1(PO_4)$ band attributable to CHA⁴⁰ could be determined in the Raman spectra of the samples prepared using 5, 15 and 30 immersing and annealing procedures. The Raman spectra of the CHA-TCP specimens are shown in Fig. 4. So, the formation of amorphous $Ca_{10}(PO_4)_6(OH)_2$ phase along with crystalline calcium phosphates is also possible.⁴¹

<u>Figure 4.</u> Raman spectra of the Ca-O-P gel samples annealed at 1000°C after each dipping procedure for 5 h in air.



Typical AFM 3D images of the calcium phosphate/hydroxyapatite thin films prepared with different number of coating procedures are presented in <u>Figs. 5-8</u>. AFM images reveal a substantial difference of their surface morphology.

<u>Figure 5.</u> Surface morphology of film (1 layer) obtained by calcination Ca-O-P gel.

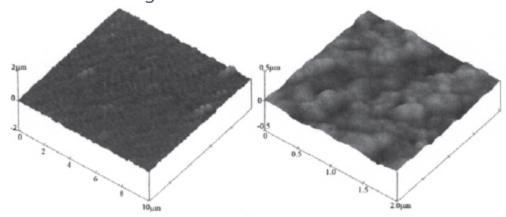


Figure 6. Surface morphology of film (5 layers) obtained by calcination Ca-O-P gel.

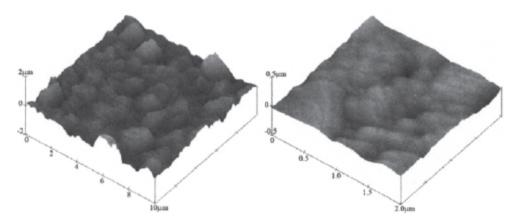
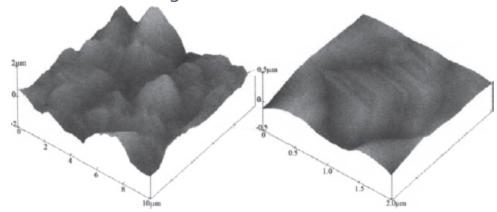
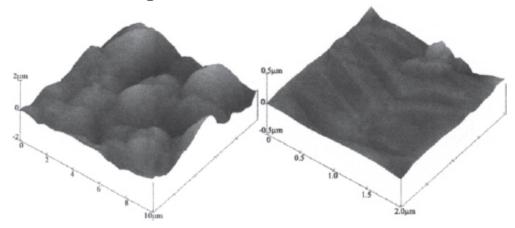


Figure 7. Surface morphology of film (15 layers) obtained by calcination Ca-O-P gel.



<u>Figure 8.</u> Surface morphology of film (30 layers) obtained by calcination Ca-O-P gel.



The surface of 1 layer film (see <u>Fig. 5</u>) exhibits smooth and homogeneous surface morphology with no special surface features. Only few submicroscopic bumps of about 250 nm diameter are visible. The intensity and size of bumps on the